## In Vacuo XPS Study of Al<sub>2</sub>O<sub>3</sub> Atomic Layer Deposition on GaN

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Vertical GaN-based switching transistors are currently receiving increased attention as an alternative to Si and SiC-based devices. GaN-based metal insulator semiconductor field effect transistors are among the most promising vertical power switching transistor architectures.[1] For secure typically OFF operation, a broad gate positive voltage span and a stable threshold voltage, reliable gate insulators are needed. For this reason, an Atomic Layer Deposited (ALD) Al<sub>2</sub>O<sub>3</sub> layer with a relatively high permittivity, large band gap, and high breakdown electric field is often used as a gate insulator. Jackson *et al.*[2] demonstrated that different wet surface treatments of GaN prior to Al<sub>2</sub>O<sub>3</sub> ALD influence the chemical composition and affect the electrical device performance. In this work, *in vacuo* X-ray Photoelectron Spectroscopy (XPS) is used to study the GaN surface composition after plasma pre-treatment and after the initial ALD cycles for both thermal (using water as precursor) and plasma-enhanced (using O<sub>2</sub>-plasma) ALD growth of Al<sub>2</sub>O<sub>3</sub> onto GaN, without exposing the sample to air.

First, the effect of a plasma treatment prior to ALD is investigated. Both NH<sub>3</sub>- and H<sub>2</sub>-plasma pretreatment can successfully remove spurious C-contamination from the pristine GaN surface while a heat treatment at 400 °C alone is not sufficient (Fig. 1). In addition, only NH<sub>3</sub>-plasma is found to remove a large fraction of the O-species that are present on the pristine GaN surface. After the first ALD-half cycle i.e. after exposure to trimethylaluminium (TMA), the signal of the Al2p and C1s XPS spectra are found to increase which is consistent with the adsorption of TMA to the surface (data not shown here). This observation demonstrates the in vacuo XPS method is sensitive to detect the growth of a signle (sub)monolayer of TMA on the surface. During the second ALD half-cycle, the TMA treated surface is exposed to water (thermal ALD) or O<sub>2</sub>-plasma (plasma-enhanced ALD process). In both cases, the data (Fig. 2) is consistent with the growth of Al<sub>2</sub>O<sub>3</sub>. Water exposure is found to offer insufficient reactivity to remove all of the adsorbed TMA ligands from the surface, while O<sub>2</sub>-plasma removes all detectable C. Further it is observed that the O concentration in the near-surface region increases dramatically after O<sub>2</sub>-plasma exposure, coinciding with the formation of O-Al, O-Ga and NO-Ga species on the surface.

Electrical data show that the density of trap states is lower for the NH<sub>3</sub>-plasma treated samples, which seems to correlate to the enhanced removal of  $Ga_xO_y$  of the pristine GaN surface.[3] In addition, more trap states were observed for samples deposited using the thermal ALD process, likely related to the presence of C-impurities, caused by the incomplete removal of the TMA ligands by water as detected by XPS. For the plasma-enhanced ALD process more hysteresis is observed between the forward and reverse bias stress sweep CV cycles, indicating a high content of stress indued interface traps. This might be explained by the presence of O-defects created during the O<sub>2</sub>-plasma pulse. The best electrical data were achieved when combining a NH<sub>3</sub>-plasma pre-treatment, followed by initial Al<sub>2</sub>O<sub>3</sub> growth using thermal ALD, and then switching to plasma-enhanced ALD to thicken the Al<sub>2</sub>O<sub>3</sub> layer.

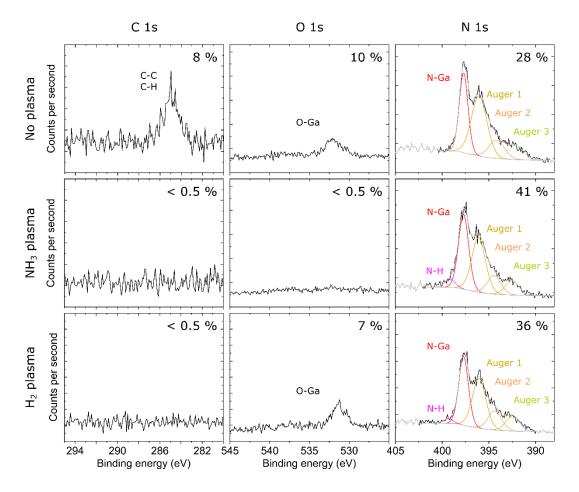
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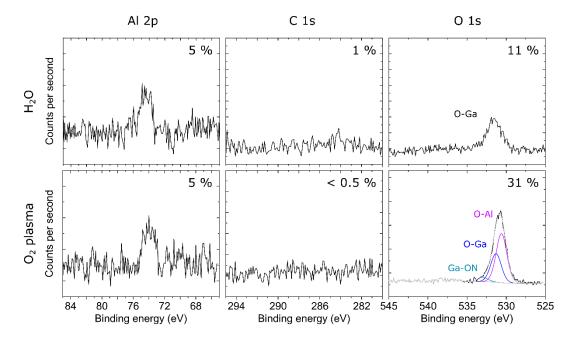
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**Figure 1** In vacuo XPS spectra acquired after heat pre-treatment and no plasma (top), NH<sub>3</sub>-plasma (middle) and H<sub>2</sub>-plasma (bottom) of a n-GaN surface.



**Figure 2** In vacuo XPS spectra acquired after water exposure (top) and O<sub>2</sub>-plasma (bottom) during respectively the thermal and plasma-enhanced  $Al_2O_3$  ALD process on a H<sub>2</sub>-plasma pre-treated GaN surface.