

# Study of metal line patterning strategy for 300 mm superconducting BEOL

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## Introduction:

The development of new Quantum technologies based on superconducting Qubits [1],[2] or spin Qubits[3], [4] becomes a major subject of interest for numerous applications in communication and data treatment. In this context, several approaches enable the fabrication of those devices [5],[6]. In this work a derivative method from VLSI device is chosen with the 28 nm FDSOI technology as a baseline. For this approach operating at low temperature, a superconducting routing development becomes crucial especially for short distance connections with specific architectures (Fig 1).

Although superconducting material properties are the key targets, these layers also have to be easily integrated in an industrial process flow. Indeed, the selected materials, commonly used in microelectronics, must keep their superconducting properties for narrow lines down to a CD of 100 nm after etching, stripping, cleaning and CMP steps.

## 1 Superconducting characterisation on blanket wafers

Metallic layers material have been selected based on their critical temperature  $T_c$  (higher than 1 K) and their integration capabilities, which led us to consider TaN and TiN. The first step was to characterize their  $T_c$  on blanket wafers. All samples resistance versus temperature were measured down to 350 mK. In a previous work, the superconducting properties of a 40 nm thick PVD TaN film were improved by adjusting the  $N_2$  flow during deposition to obtain specific phases [7]. This process development lead to an increase of the  $T_c$  up to 2,1 K compared to the standard process with a  $T_c$  of 1K (Fig 2). Some recent improvements resulted in a higher  $T_c$  of 2,6 K for the same thickness.

For the TiN, the resistance versus temperature is plotted for 3 layers from 10 nm to 40 nm thick and a critical temperature up to 3,6 K is achieved for the 40 nm layer (Fig 3). This is in good agreement with the state of the art [8]. The superconducting properties measured for both materials (TiN and TaN) are compatible with BEOL requirement for quantum application, we will then observe the influence of the patterning on the  $T_c$ .

## 2 Line patterns etching and characterisation

After TiN and TaN layer deposition, the lithography was performed on a 193nm stepper to reach 100 nm line critical dimension. Etching trials were carried out on a 300 mm industrial ICP chamber using  $Cl_2$  chemistry with or without HBr or  $CH_4$  addition. A parametric study of  $Cl_2$  based chemistries was performed to evaluate such criteria as CD profile control, and selectivity between TaN and TiN. Optical Emission Spectroscopy (OES) and Xray Photo-electron Spectroscopy (XPS) were conducted in order to understand the etching mechanisms.

Regarding the TiN, straight profiles (around  $85^\circ$  of steepness) and good CD control (CD bias = 6 nm) are achieved with  $Cl_2$ /HBr chemistries but micro masking is observed (Fig 4). The micromasking phenomenon will be further discussed in terms of etching mechanisms. The use of  $CH_4$  instead of HBr overcomes this issue while keeping a good profile (around  $81^\circ$  of steepness) with an etch rate of 70 nm/min (Fig 5). Regarding TaN, lower etch rates are observed in comparison with TiN which might be attributed to a higher Ta-N binding energy and lower by-product volatility (as example 60 nm/min versus 125 nm/min with  $Cl_2$ /Ar). A selectivity TiN:TaN of about 4.6 is achieved by adding  $CH_4$  in the  $Cl_2$  based chemistry with an etch rate of 70 nm/min for TiN and 18 nm/min for TaN. Interestingly, the steepness can be controlled from  $74^\circ$  with  $Cl_2$ /HBr (Fig 6) to  $60^\circ$  with pure  $BCl_3$  by tuning the etching chemistry.

The  $T_c$  of TiN lines etched with  $Cl_2$ / $CH_4$  and TaN lines etched with  $Cl_2$ /Ar were measured. The results show only a minor decrease of  $T_c$  (-0.2 K) and a small  $\Delta T$  increase of the TaN lines compared to the blanket results (Fig 7). However, the TiN lines exhibit multiple transitions (Fig 8), which might be explained by an inhomogeneity of the line. We will further discuss those unexpected  $T_c$  transitions.

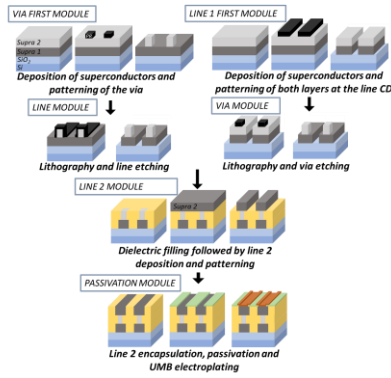


Figure 1 Patterning approach : Process Flow

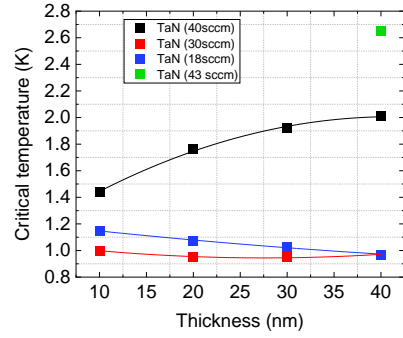


Figure 2 Critical temperature variation of TaN thin film versus thickness and Nitrogen flow, the lines are here to guide the eye

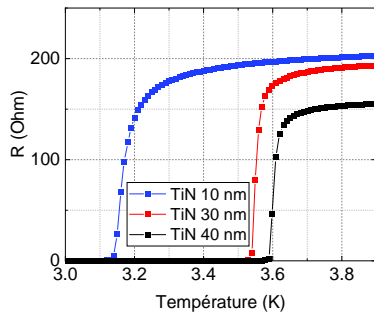


Figure 3 Resistance as a function of the temperature for TiN samples with thicknesses from 10 to 40 nm

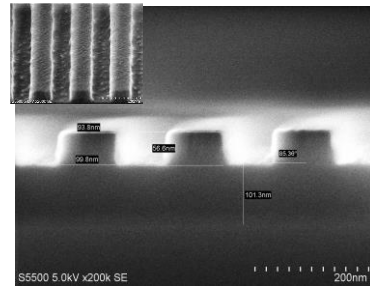


Figure 4 Cross section of 100 nm TiN lines etched with Cl<sub>2</sub>/HBr

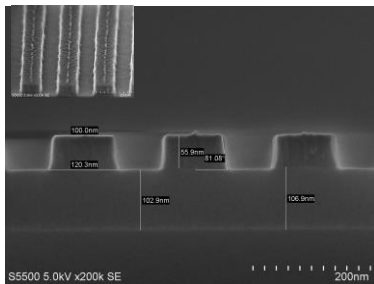


Figure 5 Cross section of 100 nm TiN lines etched with Cl<sub>2</sub>/CH<sub>4</sub>

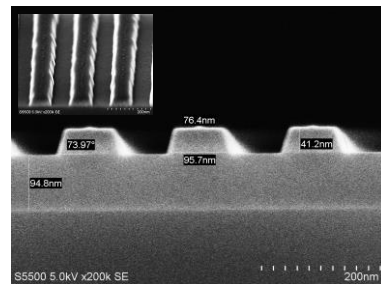


Figure 6 Cross section of 100 nm TaN lines etched with Cl<sub>2</sub>/HBr

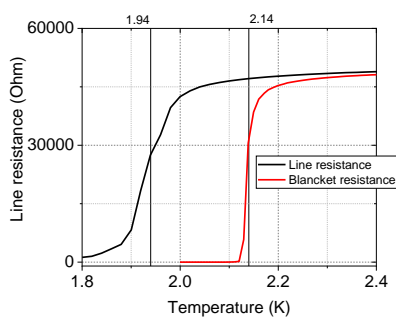


Figure 7 The resistance as a function of temperature for a 100 nm CD TaN line with a length of 60 μm and a blanket TaN layer with a thickness of 40 nm

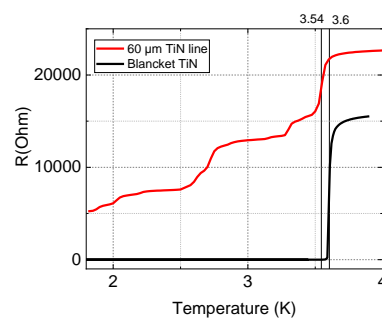


Figure 8 Resistance versus temperature of a CD 100 nm TiN line of 60 μm with a thickness of 40 nm and a blanket TiN layer

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