

# STUDY OF MBE GROWTH AND THERMAL STABILITY OF $\text{Bi}_2\text{MnTe}_4$ THIN FILMS

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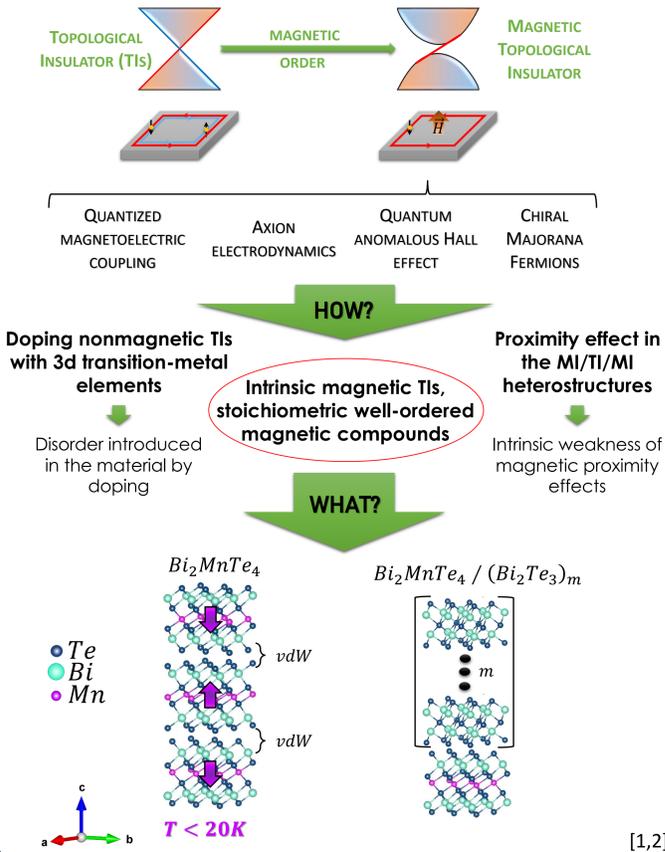
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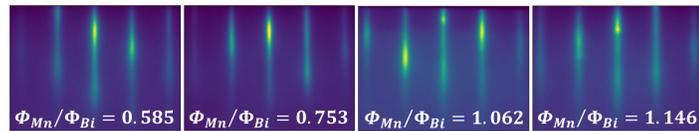
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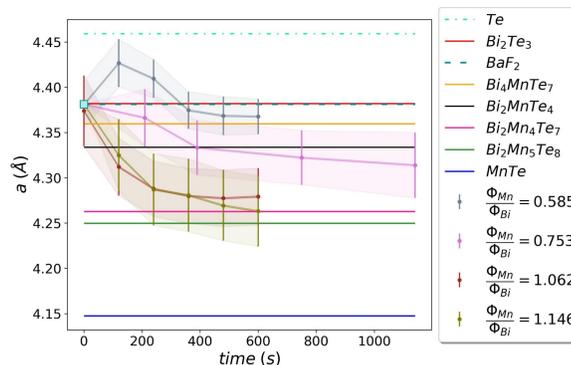
## INTRODUCTION



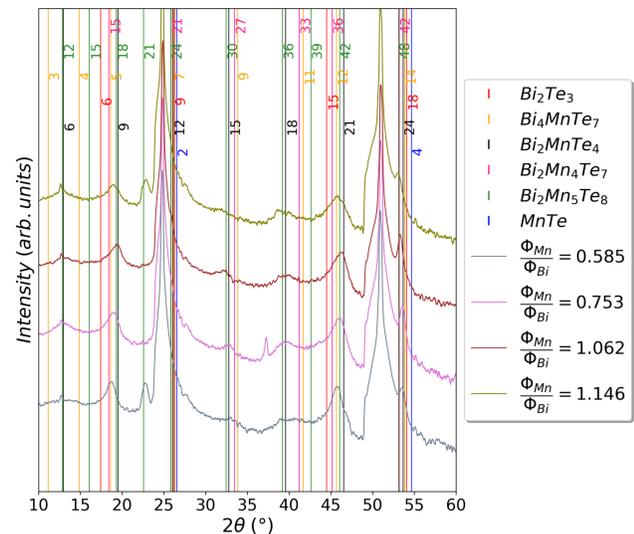
## Mn:Bi FLUX RATIOS EFFECT ON THE FILMS CRYSTAL STRUCTURE



RHEED pattern of each film after growth evidencing the good quality of films surface



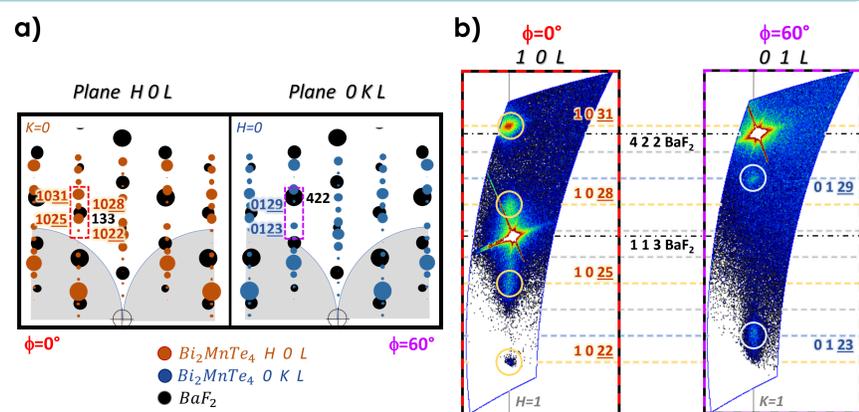
In-plane a lattice constant evolution at the beginning of the growth. Since we use a relative measurement method, the starting value represented with a square, which corresponds to the substrate, is fixed all the films. As the ratio Mn:Bi grows, the in-plane lattice constant moves to values that correspond to Mn-rich phases. The bump observed for the Mn:Bi=0.585 results from an excess of Te.



XRD  $\theta$ - $2\theta$  characterization of the films. Peak positions result from the interplay of the different phases. In Mn:Bi=0.585, the main phase is  $\text{Bi}_4\text{MnTe}_7$ . As Mn:Bi rises,  $\text{Bi}_2\text{MnTe}_4$  becomes in the main phase. However, spurious  $\text{Bi}_2\text{Mn}_4\text{Te}_7$  and  $\text{Bi}_2\text{Mn}_5\text{Te}_8$  phases emerge parallelly. In Mn:Bi=1.146 a major contribution of  $\text{Bi}_2\text{Mn}_5\text{Te}_8$  is evidenced.

## TWIN DOMAINS DEFECT STUDY

**a)** Simulation of the expected reflections for the reciprocal planes H 0 L and O K L of  $\text{Bi}_2\text{MnTe}_4$ . The relative intensity is represented by the circle size. **b)** Reciprocal space mapping (RSM) of the sample Mn:Bi=1.062 for indicated reciprocal regions. The absence of an overlapping of the two reciprocal plane patterns evidences the absence of twin domain defects in the films. Even more, a good agreement in the expected relative intensities is observed.



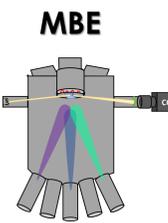
Diagonal modulation in the RHEED pattern also reveals the absence of twin defects [7].

## METHODS

Films of  $\text{Bi}_2\text{MnTe}_4$  were grown under co-evaporation of Bi, Mn and Te on  $\text{BaF}_2$  (111) by molecular beam epitaxy (MBE). For all the samples, the Te:Bi ratio was fixed to  $3.01 \pm 0.05$  and the  $T_{\text{sub}}=380^\circ\text{C}$ .

The films were studied using *in-situ* reflection high energy electron diffraction (RHEED) and *ex-situ* X-ray diffraction (XRD).

The different  $\text{Bi}_m\text{Mn}_n\text{Te}_z$  phases present in this work are the following:



	$\Phi_{\text{Mn}}/\Phi_{\text{Bi}}$					
	0					$\infty$
Stacking Pattern						
Chemical formula	$\text{Bi}_2\text{Te}_3$	$\text{Bi}_4\text{MnTe}_7$ [*]	$\text{Bi}_2\text{MnTe}_4$	$\text{Bi}_2\text{Mn}_4\text{Te}_7$	$\text{Bi}_2\text{Mn}_5\text{Te}_8$	$\text{MnTe}$
# block per unit cell	3	1	3	3	3	1
c	30.510 [3]	23.818 [4]	40.950 [**]	72.251 [***]	82.686 [***]	6.710 [6]
a	4.382 [3]	4.360 [4]	4.334 [2]	4.263 [***]	4.250 [***]	4.148 [6]

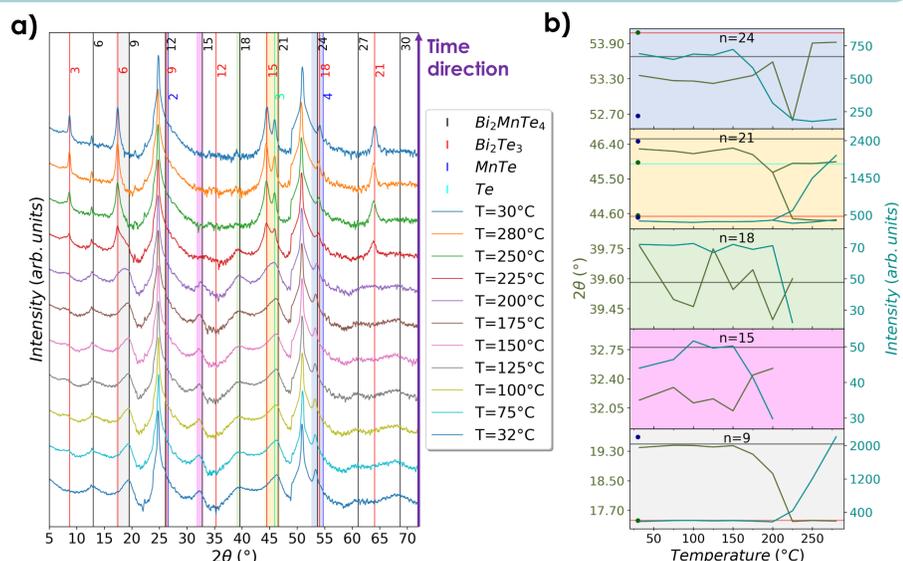
\* Compact notation of  $\text{Bi}_m\text{Mn}_n\text{Te}_z/\text{Bi}_2\text{Te}_3$   
 \*\* Calculated based on XRD measurements performed on recently grown films.  
 \*\*\* Estimated based on the  $\text{Bi}_2\text{MnTe}_4$ ,  $\text{Bi}_2\text{Te}_3$  and MnTe data.  $\text{Bi}_2\text{Mn}_4\text{Te}_7$  is reported in [5].

## ACKNOWLEDGEMENTS

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## THERMAL STABILITY OF FILMS STUDY

**a)** XRD  $\theta$ - $2\theta$  spectrums of the sample Mn:Bi=1.062 as a function of temperature, acquired in rough vacuum conditions ( $\approx 10^{-1}$  mbar). **b)** Evolution of the peak's position and peak's intensity for reflections L= 9, 15, 18, 21. The dot represents the peak position after cooling down. Over  $150^\circ\text{C}$ , the study shows that the film decomposes in a mixture of  $\text{Bi}_2\text{Te}_3$  and metallic Te, in which the Mn seems to form an amorphous phase (from additional analysis). Since the film does not recover the initial state after cooling down, the transformation is irreversible. The observations are partially in agreement with previous studies in bulk crystals [2].



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