



# Towards large-scale hexagonal boron nitride 2D layers: a chemical approach

**Catherine Journet**  
*Bérangère Toury*



*Laboratoire des  
Multimatériaux & Interfaces  
Lyon University*



*Yangdi Li*

*Vincent Garnier*  
*Philippe Steyer*



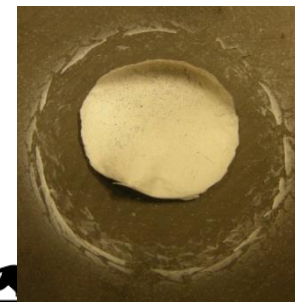
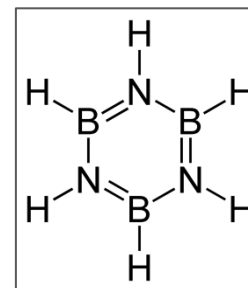
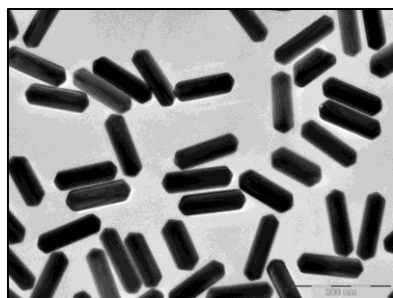
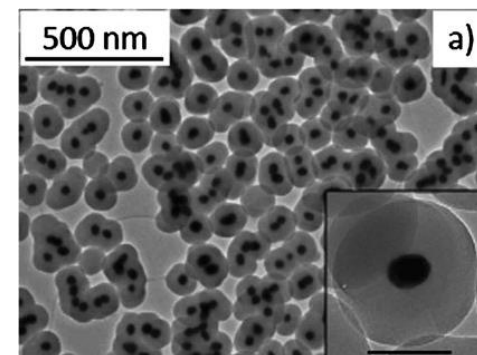
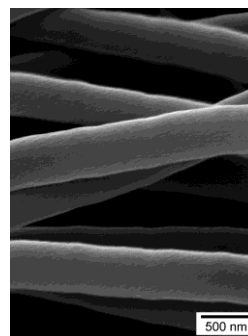
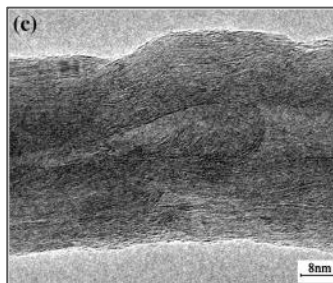
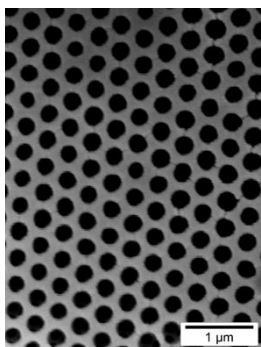
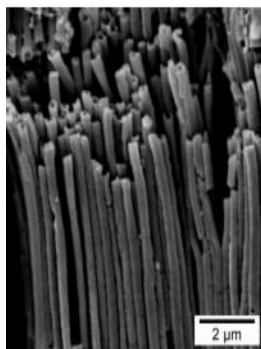
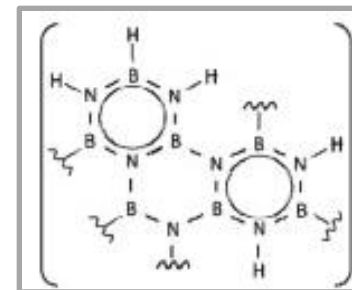
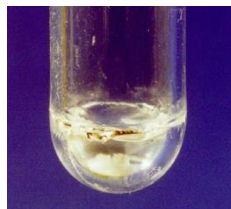
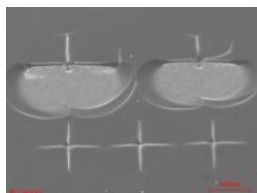
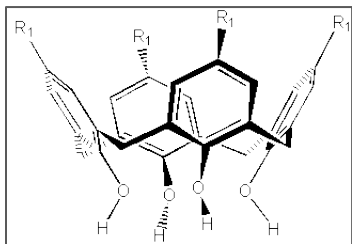
*Laboratoire Matériaux  
Ingénierie et Science  
INSA Lyon*



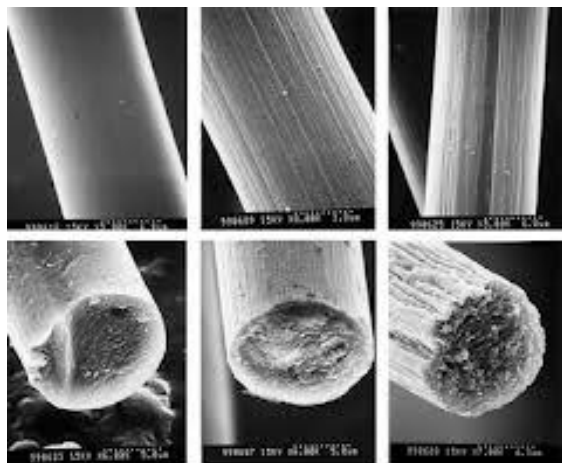
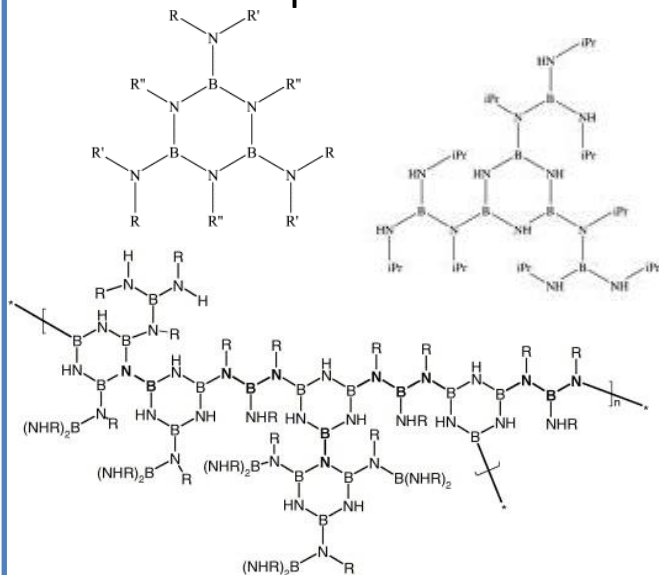
<http://lmi.cnrs.fr>

from molecule ...

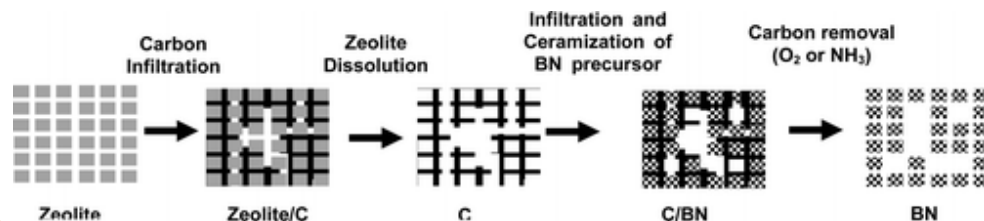
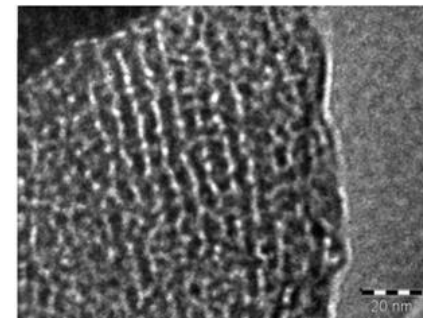
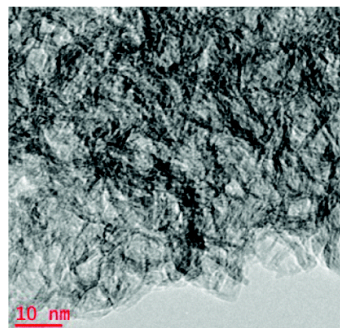
... to material



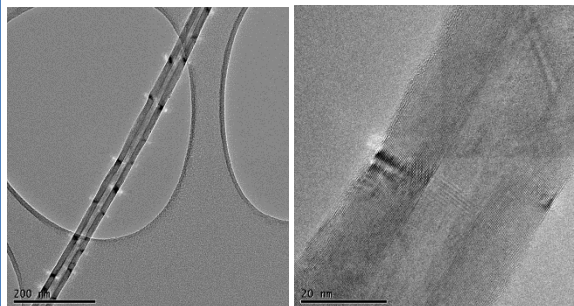
### BN fibers new precursors



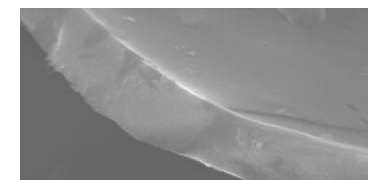
### micro- and meso-porous BN by *template* or direct synthesis



### BN nanotubes



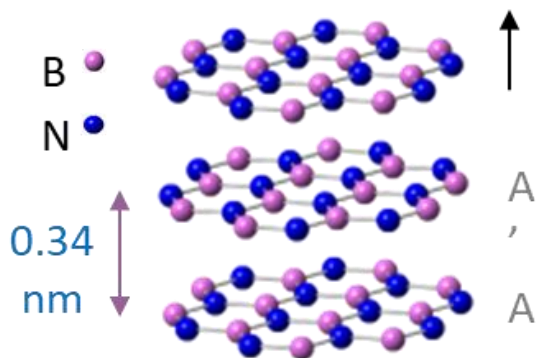
### *h*-BN coatings



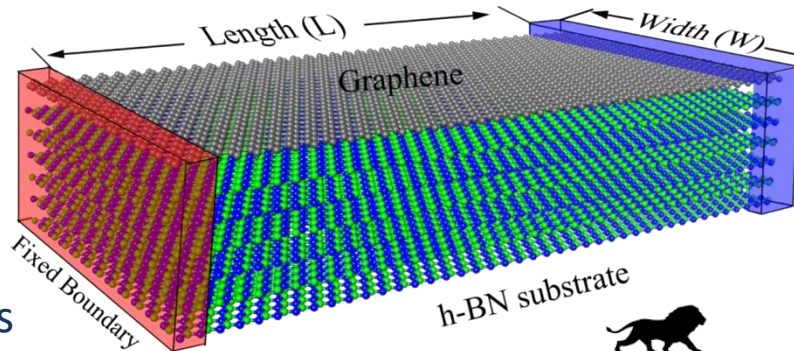
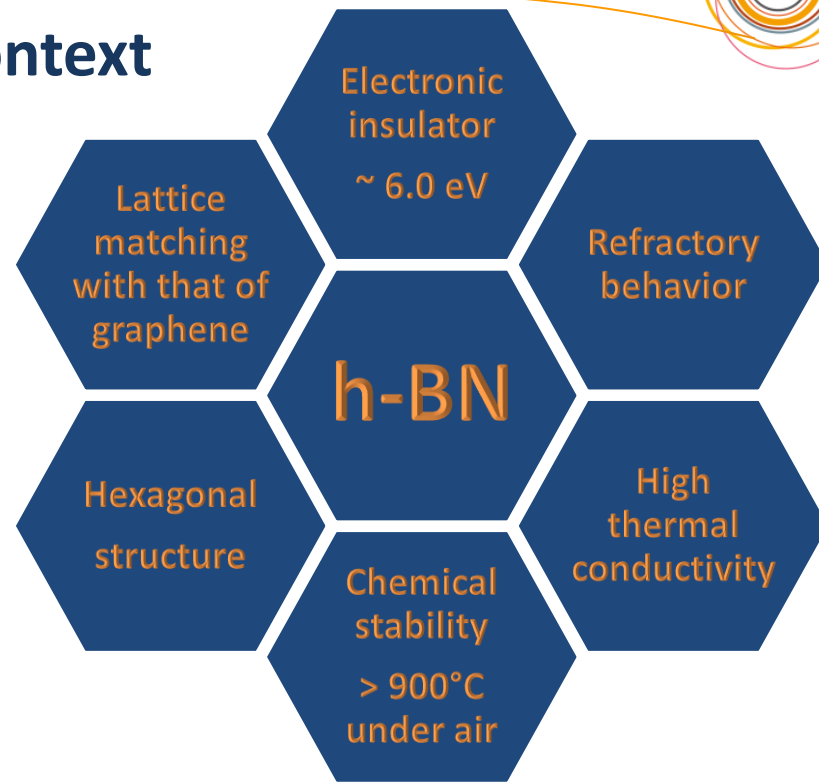
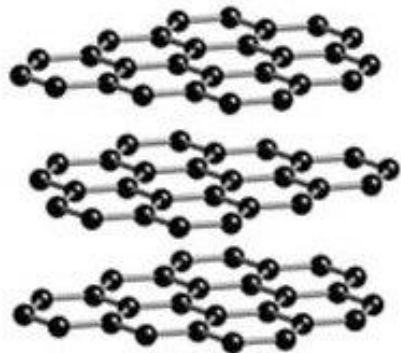


# General context

**h-BN** also called white graphite



**Graphite**



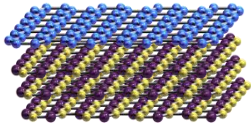
➤ Great potential for electronics and optics

Zhongwei Zhang et al 2017 Nanotechnology 28 225704

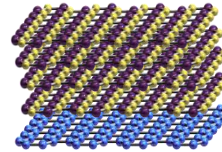


## Towards layered hBN

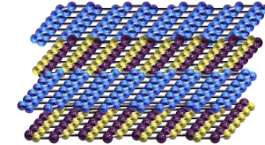
Several needs of h-BN samples :



Thick layers to be used as substrate for graphene



Encapsulating layers of graphene and other 2D materials



Dielectric layers in heterostructures

Need of both monolayers and thick layers (> 10 – 50 nm)

## Our approach

### High quality h-BN source



#### Polymer Derived Ceramics (PDCs)

- Tailored molecular precursor
- Non-oxide system for high purity
- Liquid precursors easy to handle
- Possibility for various shaping methods for specific ceramic shape

*S. Yuan, et al. Crystals. 2016, 6, 55*

*S. Yuan, et al. Nanoscale. 2014, 6, 7838-7841*

*P. Colombo. J. Am. Ceram. Soc. 2010, 93, 1805-1873*



#### Spark Plasma Sintering (SPS)

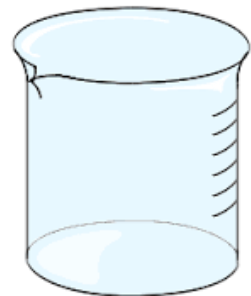
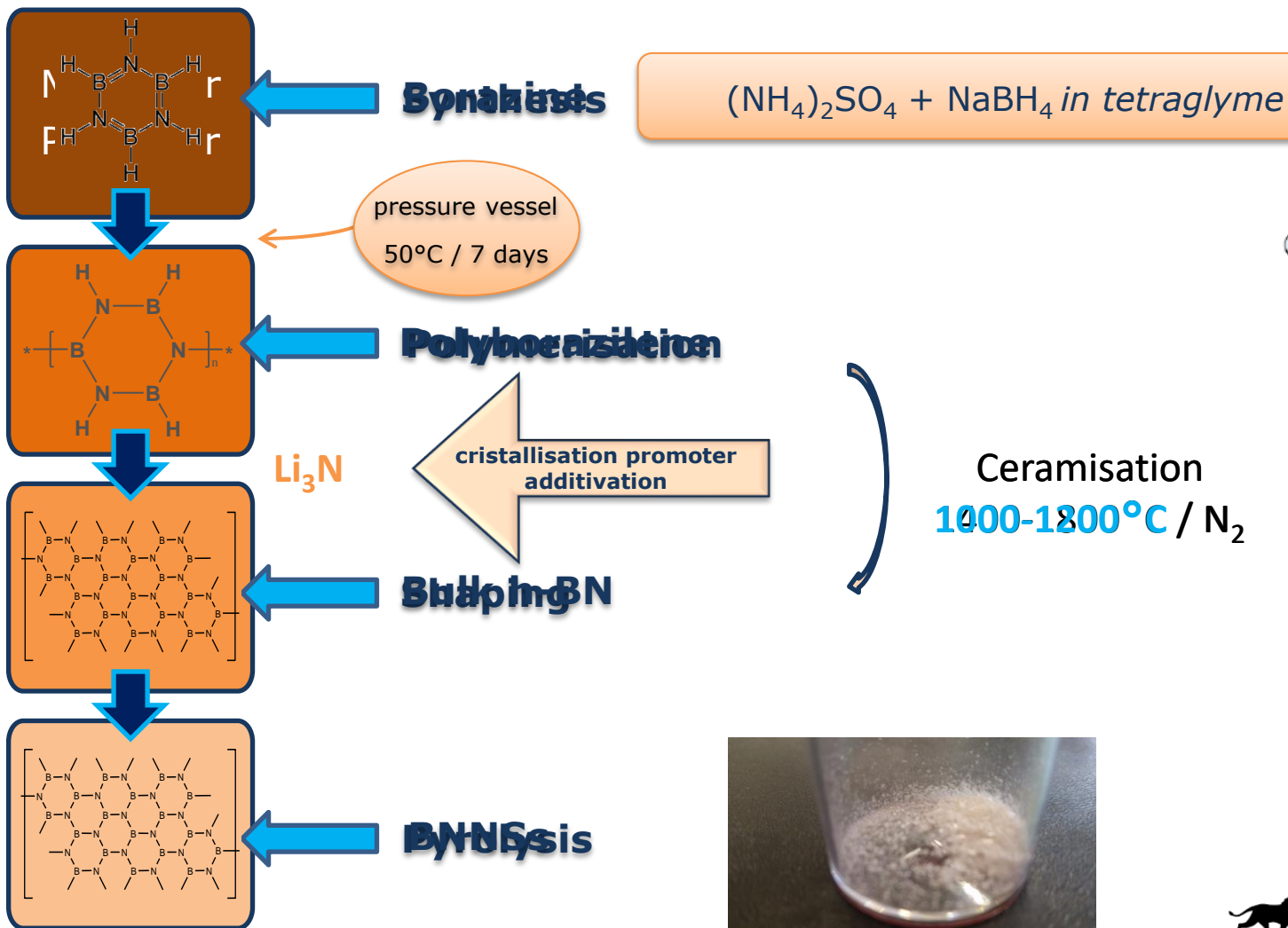
- Rapid and efficient processing method for densification on both lab scale and industrial level
- Softer condition compared with HPHT method (2100°C, 5.5GPa, 80h)

*S. Yuan, et al. Sci. Rep. 2016, 6, 20388*

*E. Bernardo, et al. Ceram. Int. 2014, 40, 14493-14494*

*K. Watanabe, et al. Nat. Mater. 2004, 3, 404-409*

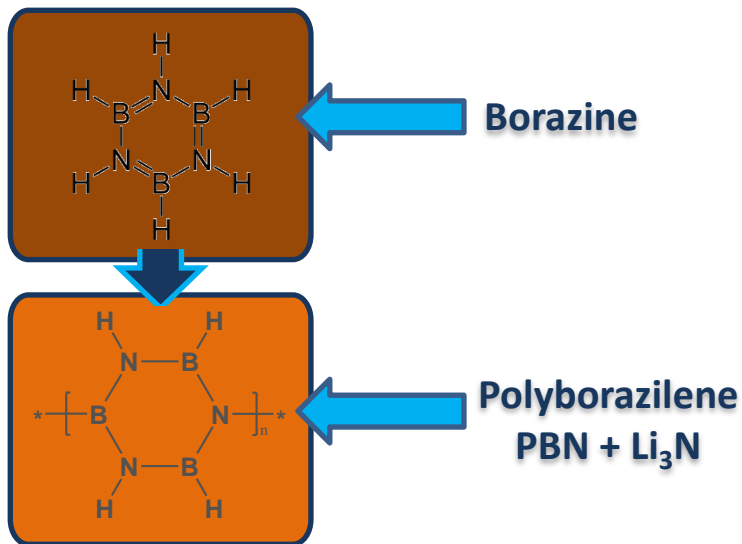
# Polymer Derived Ceramics (PDCs) route





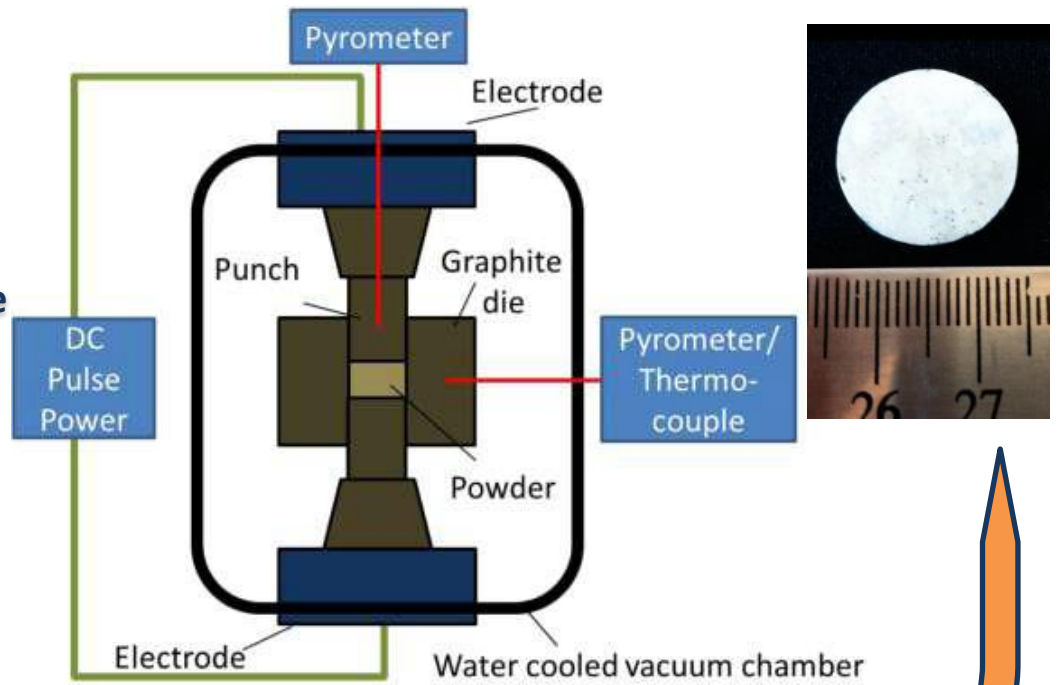
# Combination PDCs + SPS

O. Guillon, et al. *Adv. Eng. Mat.* **2014**, 16, 831



stabilized 600°C, 1h

**Powder**



## Spark Plasma Sintering (SPS)

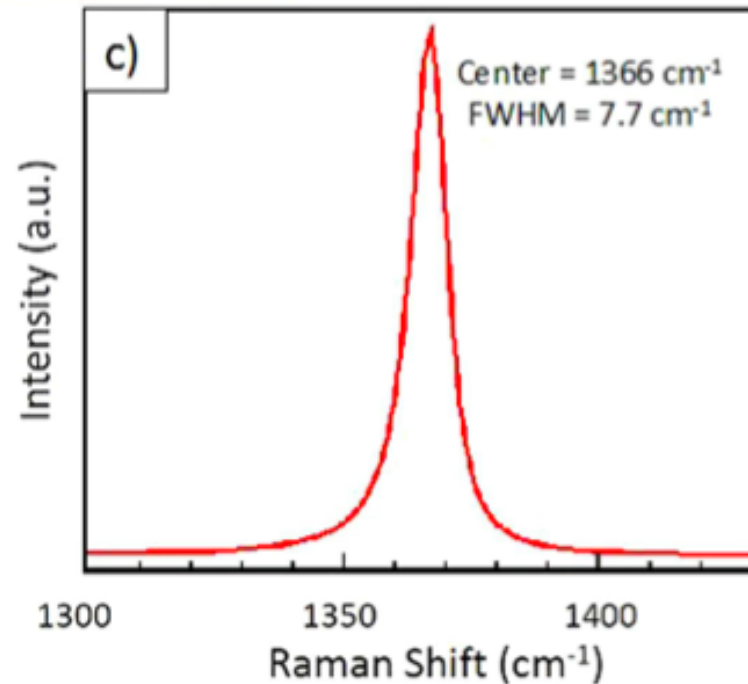
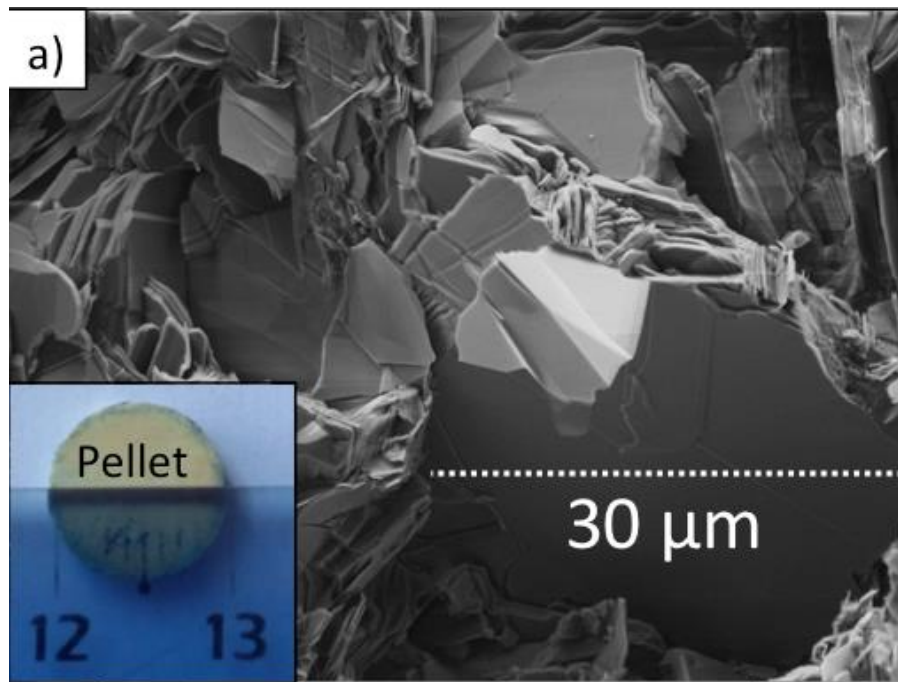
90 MPa  
770 A  
1800°C  
1h





# Characterization of the bulk material

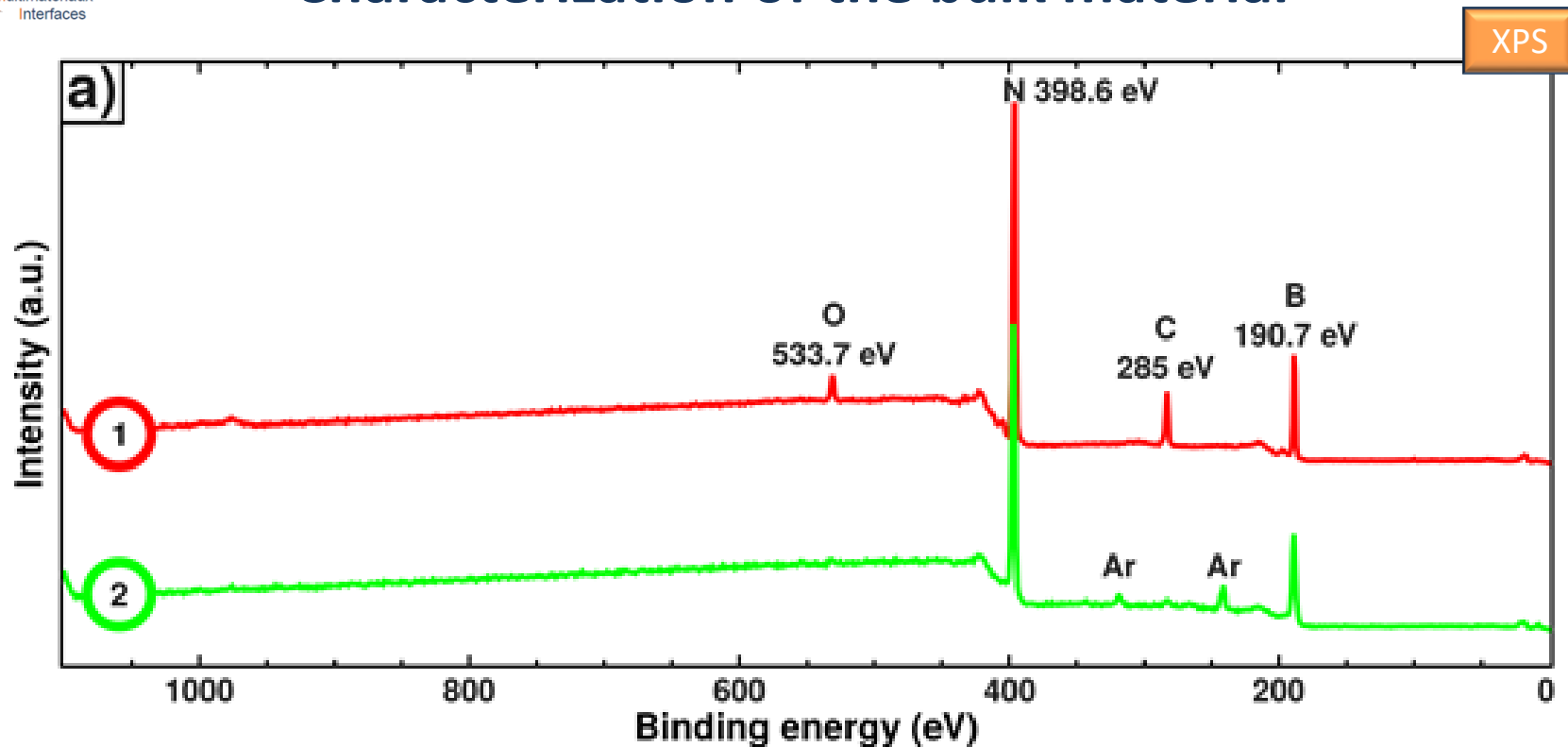
XR | EDS | SEM | S



polycrystalline sample made of single crystal flakes

S. Yuan, et al. *Sci. Rep.* **2016**, *6*, 20388

# Characterization of the bulk material



① Before abrasion :

ratio B/N = 0.97 (close to 1)  
< 0.5%at. for O1s and 1.3%at. for C1s

② After abrasion (1 $\mu$ m) :

elimination of C and O (just contaminants)  
preferential sputtering of nitrogen atoms

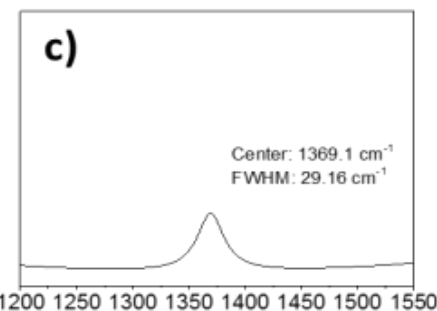
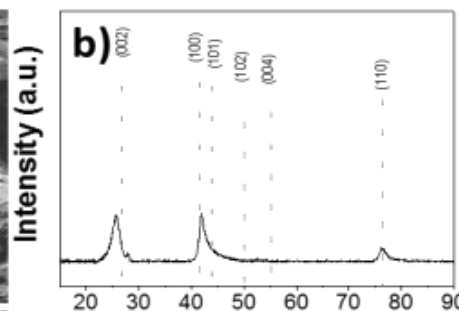
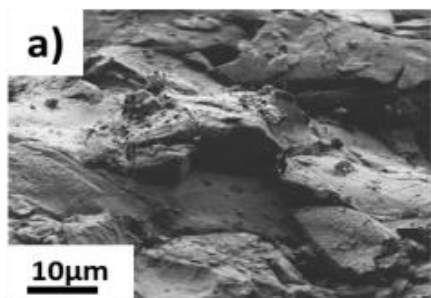


# Influence of the crystallization promoter

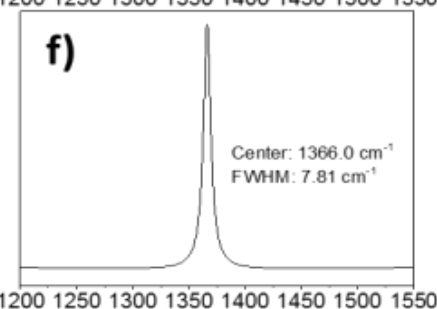
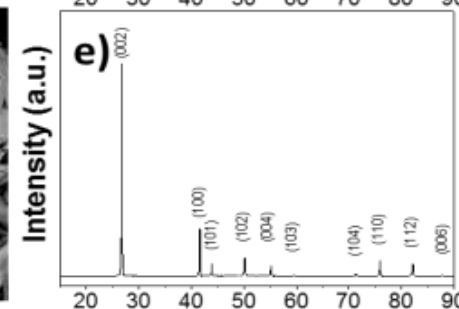
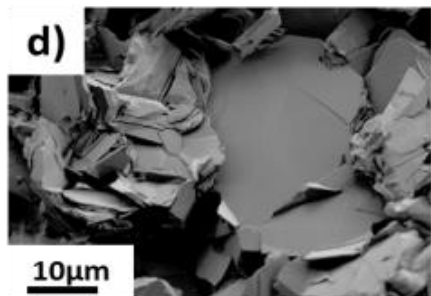
| No . | Pre-ceramic Composition | Li <sub>3</sub> N wt. % | Temperature °C | Pressure MPa | Dwelling time hour |
|------|-------------------------|-------------------------|----------------|--------------|--------------------|
| 1    | PBN                     | 0                       | 1800           | 90           | 1                  |
| 2    | PBN+Li <sub>3</sub> N   | 5                       | 1800           | 90           | 1                  |
| 3    | PBN+Li <sub>3</sub> N   | 10                      | 1800           | 90           | 1                  |

# Influence of the crystallization promoter

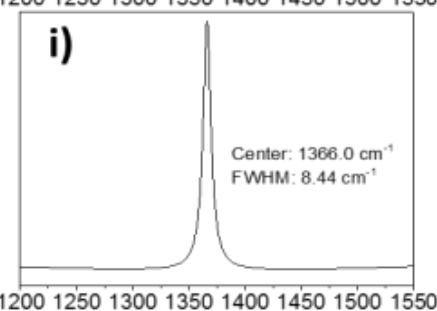
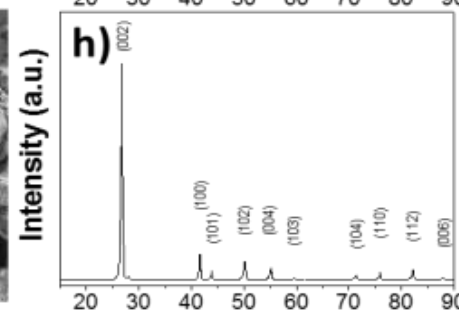
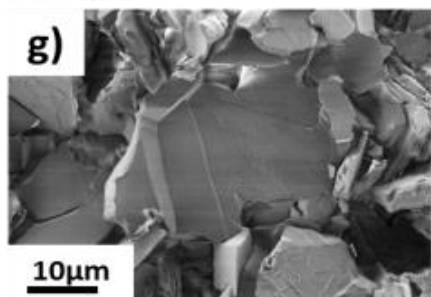
0%  $\text{Li}_3\text{N}$



5%  $\text{Li}_3\text{N}$



10%  $\text{Li}_3\text{N}$



- $\text{Li}_3\text{N}$  addition necessary for obtaining well-crystallized h-BN flakes
- No significant structural difference between 5% and 10 wt.%
- Ceramic yield decreases when increasing  $\text{Li}_3\text{N}$  amount

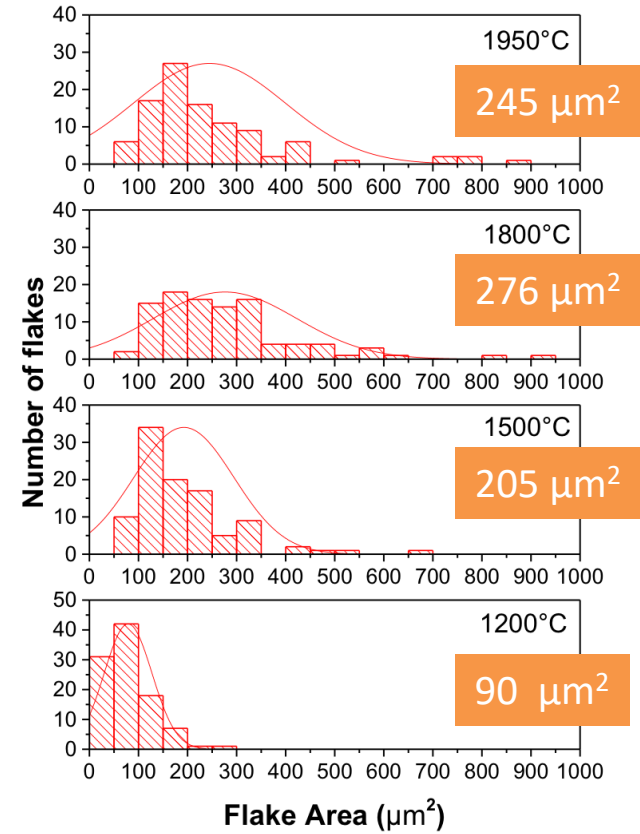
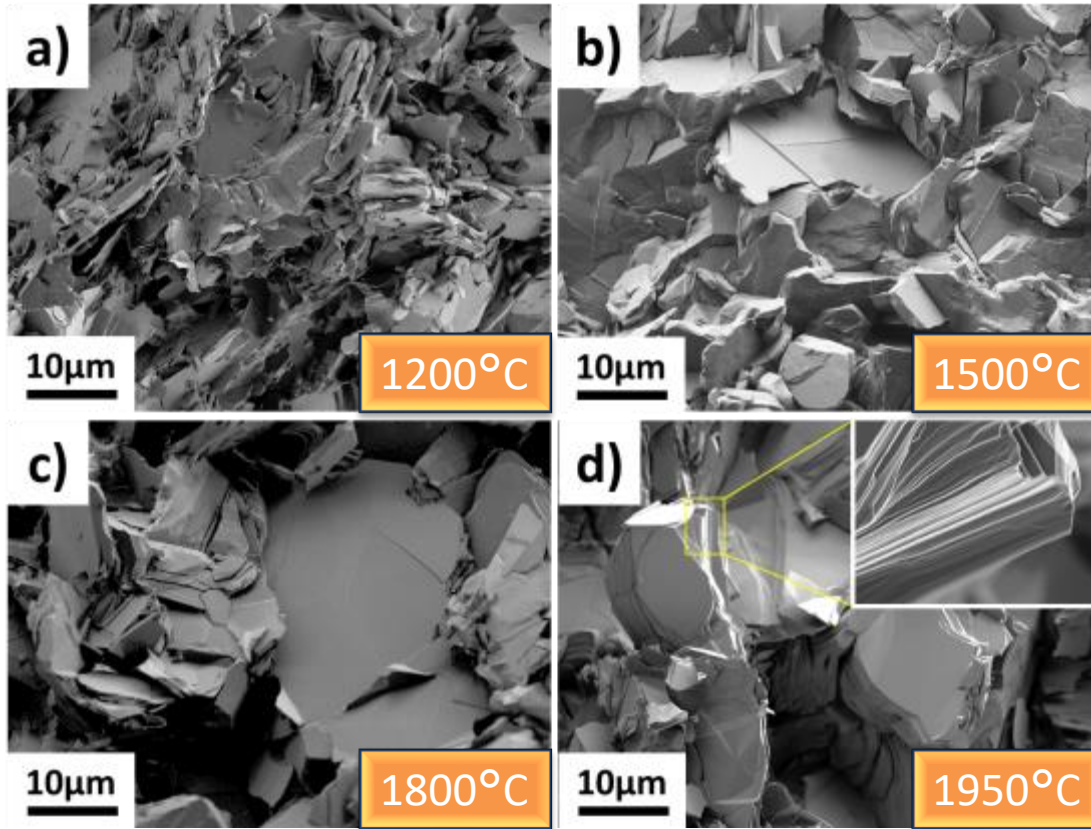


## Influence of the sintering T

| No . | Pre-ceramic Composition | Li <sub>3</sub> N wt. % | Temperature °C | Pressure MPa | Dwelling time hour |
|------|-------------------------|-------------------------|----------------|--------------|--------------------|
| 1    | PBN+Li <sub>3</sub> N   | 5                       | 1200           | 90           | 1                  |
| 2    | PBN+Li <sub>3</sub> N   | 5                       | 1500           | 90           | 1                  |
| 3    | PBN+Li <sub>3</sub> N   | 5                       | 1800           | 90           | 1                  |
| 4    | PBN+Li <sub>3</sub> N   | 5                       | 1950           | 90           | 1                  |



# Influence of the sintering T



- Flakes size  $\uparrow$  with T to reach a maximum area of 276  $\mu\text{m}^2$  at 1800°C and  $\downarrow$  at 1950°C
- When sintered off defined but elongation is a  $\uparrow$  faster flakes and growing in different directions
- decreasing the flakes size
- flakes are significantly larger

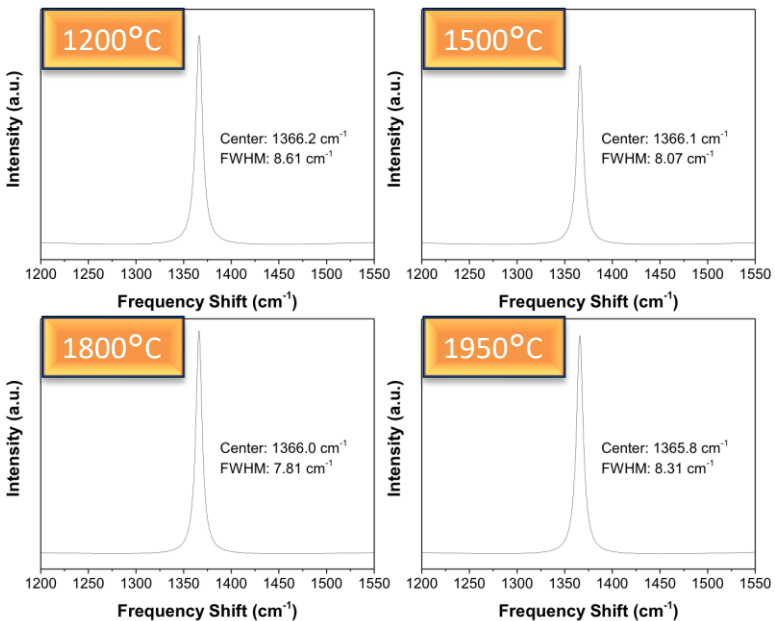
best compromise between 1500 and 1800°C





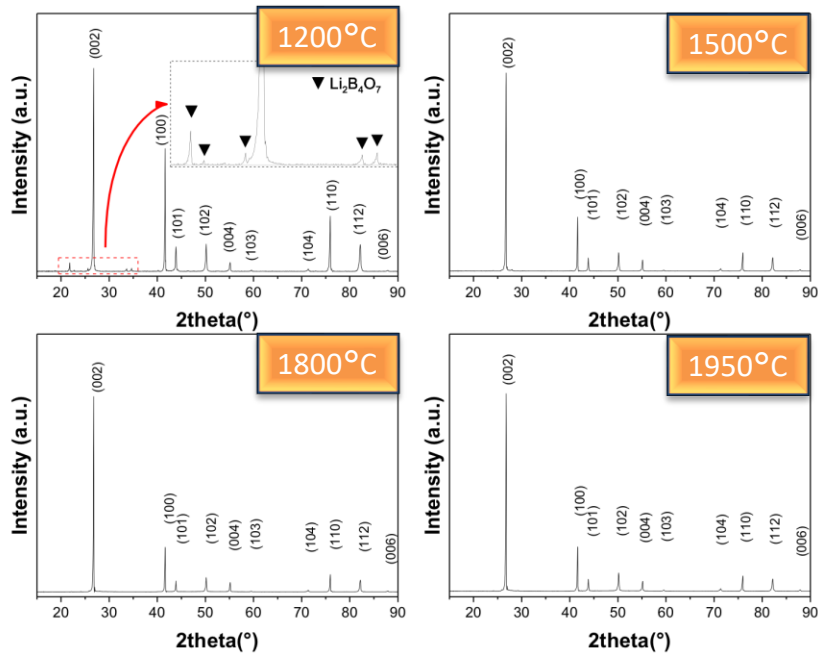
# Influence of the sintering T

Raman



Very good Raman signature with a FWHM  $\approx 8 \text{ cm}^{-1}$

XRD



Good crystalline structure of h-BN

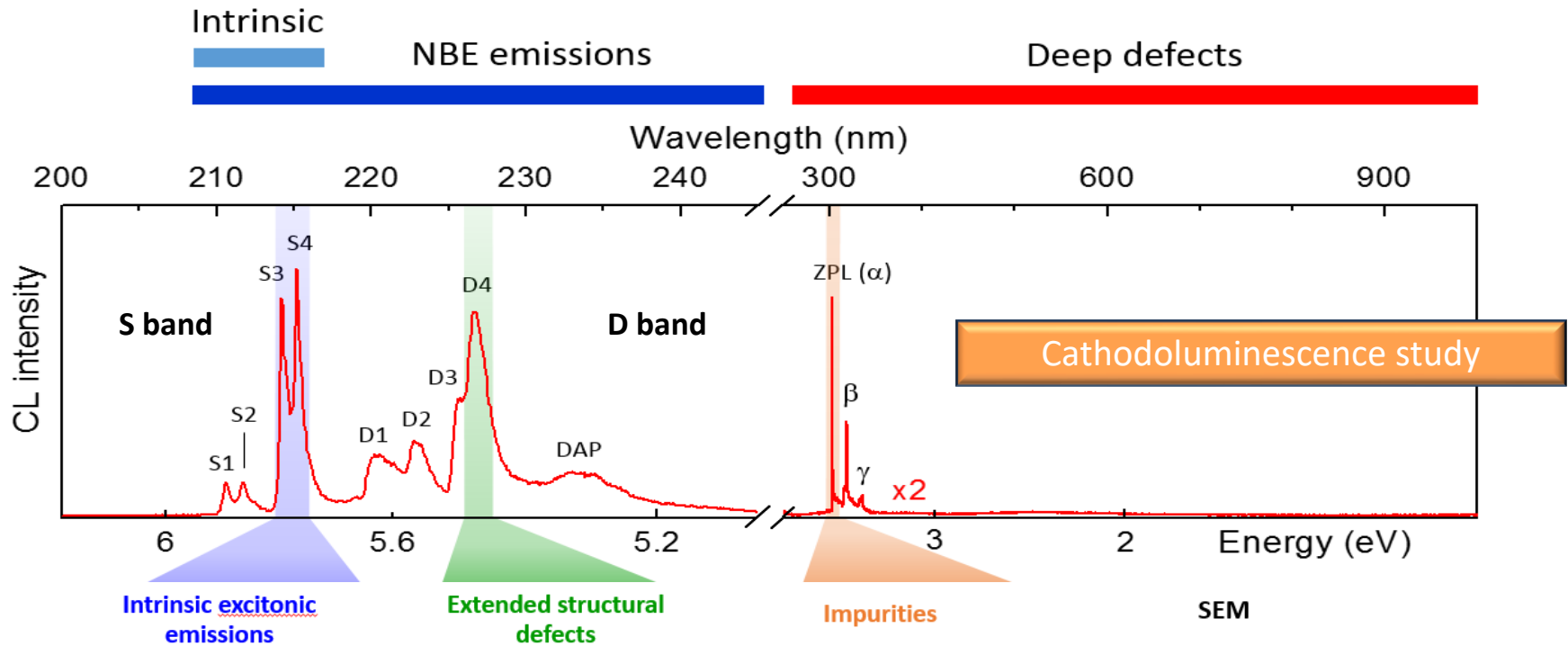


# Influence of the crystallization promoter and sintering T

A tool of interest

Collab. GEMAC & ONERA (A. Plaud, L. Schue, J. Barjon, A. Loiseau)

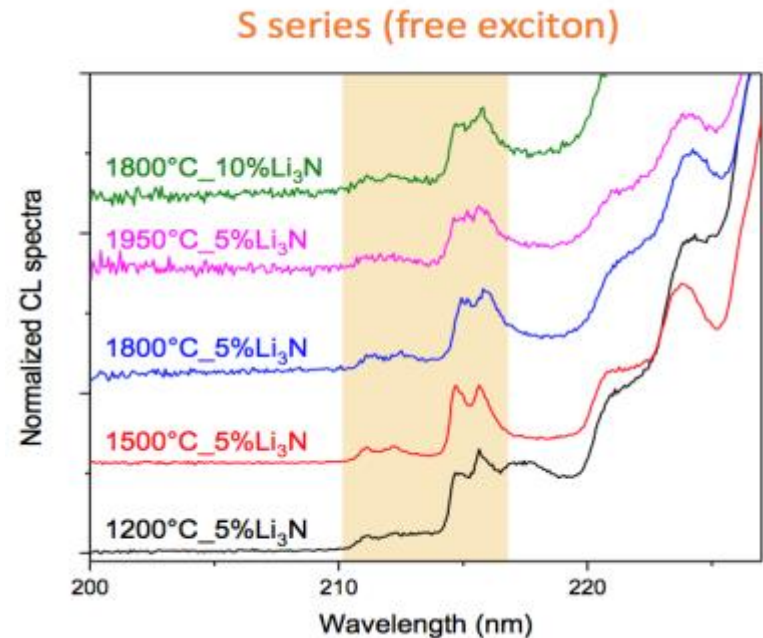
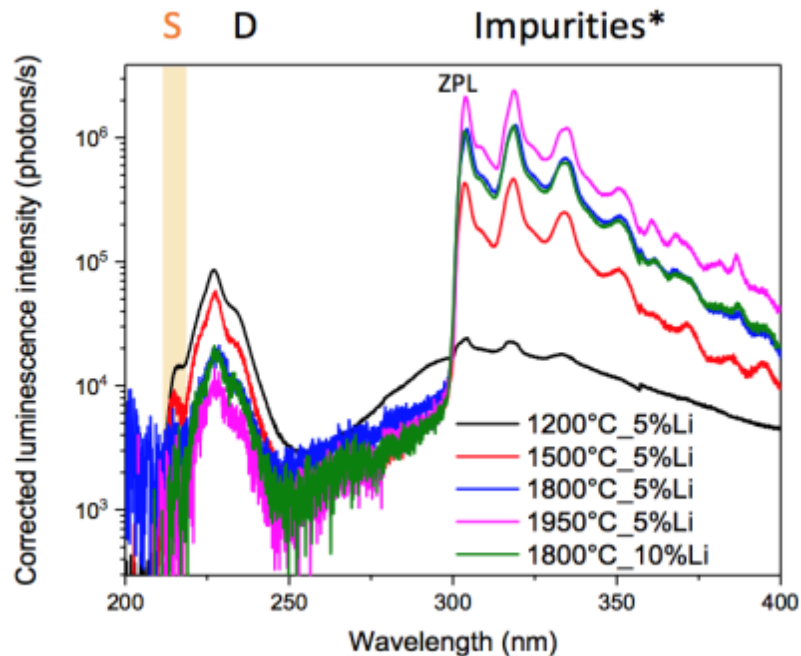
- Investigation of optical and excitonic properties
- identification of different classes of defects and their impact on optical properties





# Influence of the crystallization promoter and sintering T

Indication of the overall material quality, accounting for both purity and crystallinity



- Observation of intrinsic exciton emission (S-lines)
- Absence of defect-related emissions (D-lines)
- Presence of impurities when increasing the sintering temperature : contamination ?

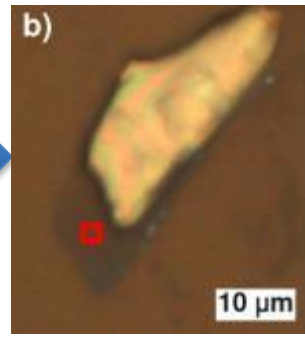
Collab. GEMAC & ONERA (L. Schue, J. Barjon, A. Loiseau)



# From h-BN pellets ... towards flakes



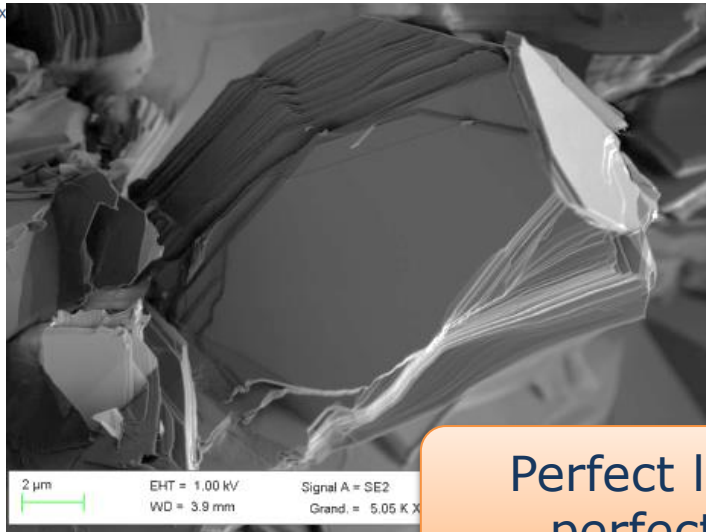
well-crystallized  
h-BN bulk sample  
(pellet)



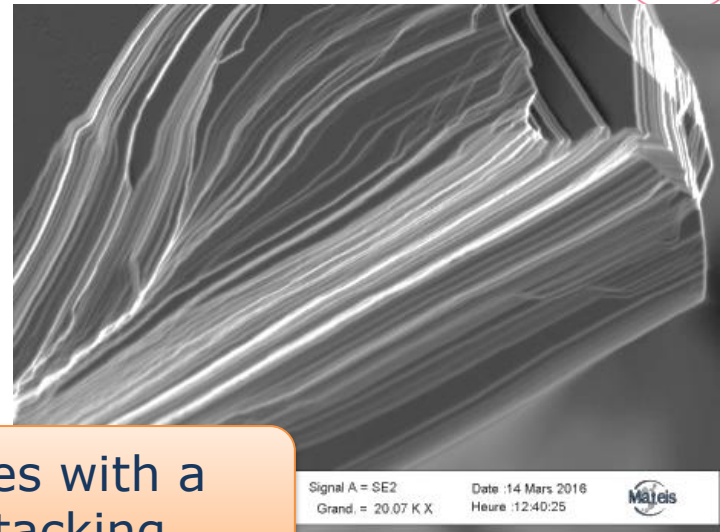
flake  
(single crystal)



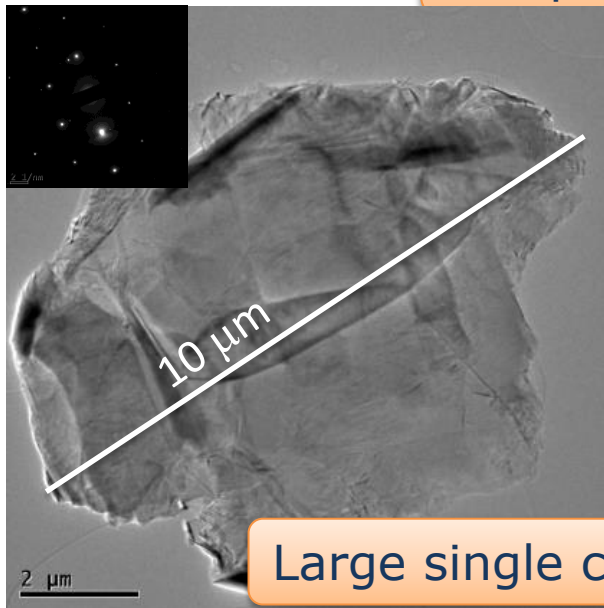
# Characterization of the flakes



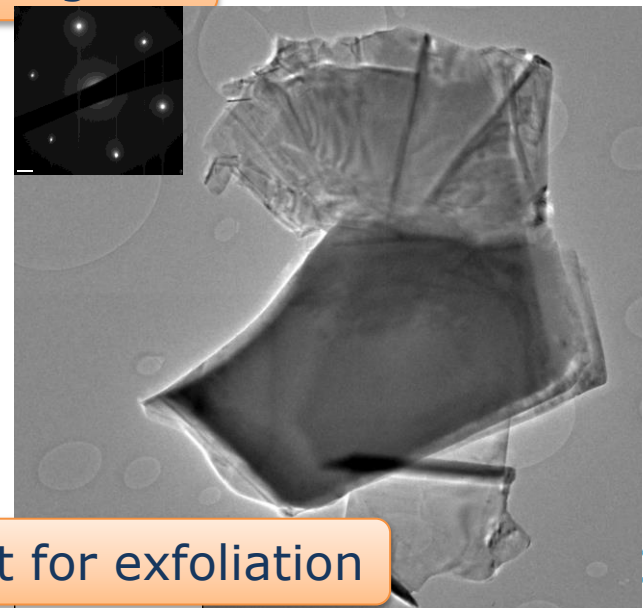
SEM



Perfect large flakes with a perfect layers stacking

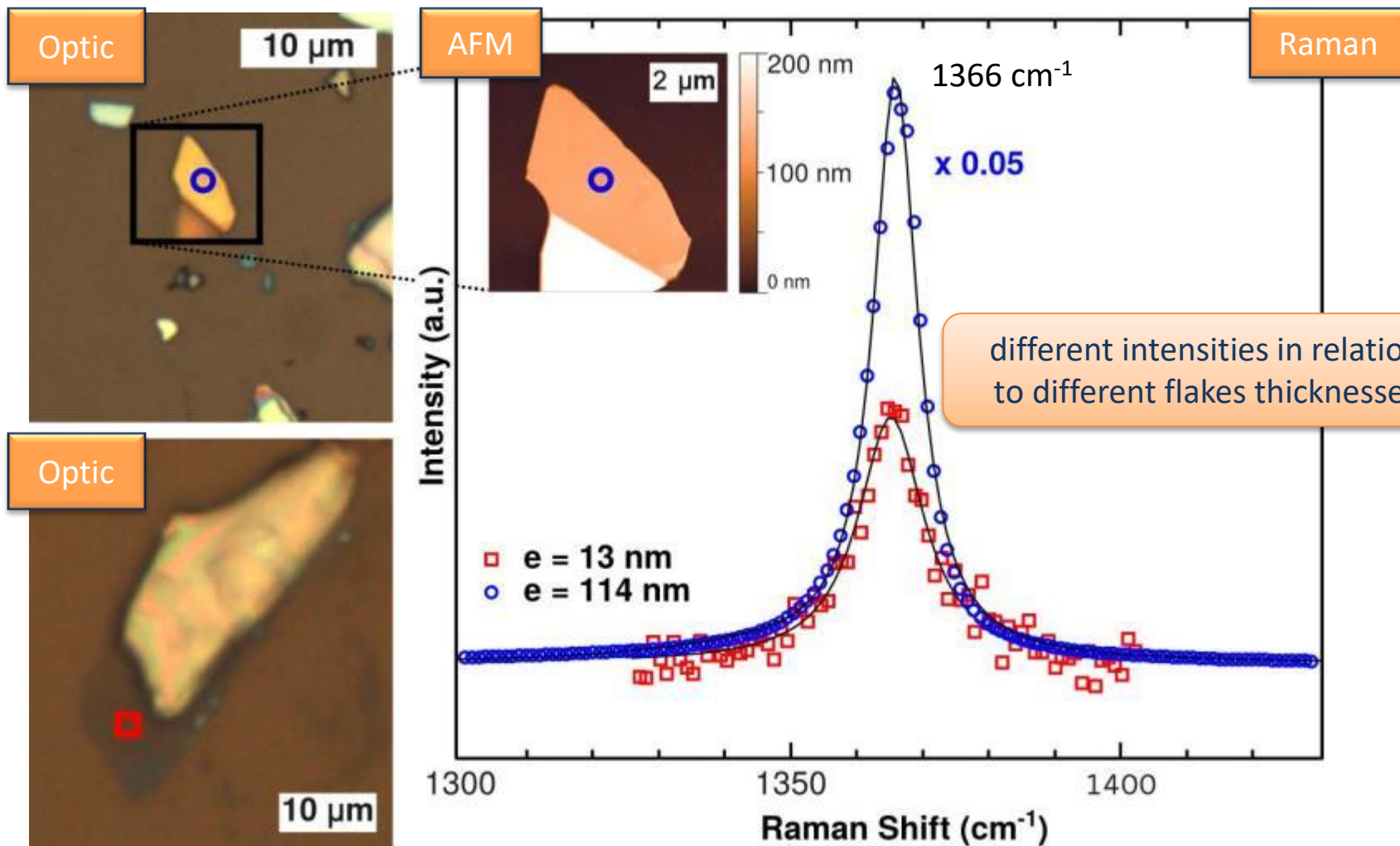


TEM



Large single crystal convenient for exfoliation

# Characterization of the flakes

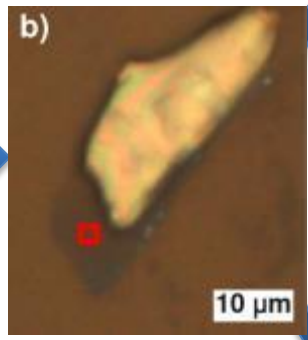


# From h-BN pellets ... towards flakes

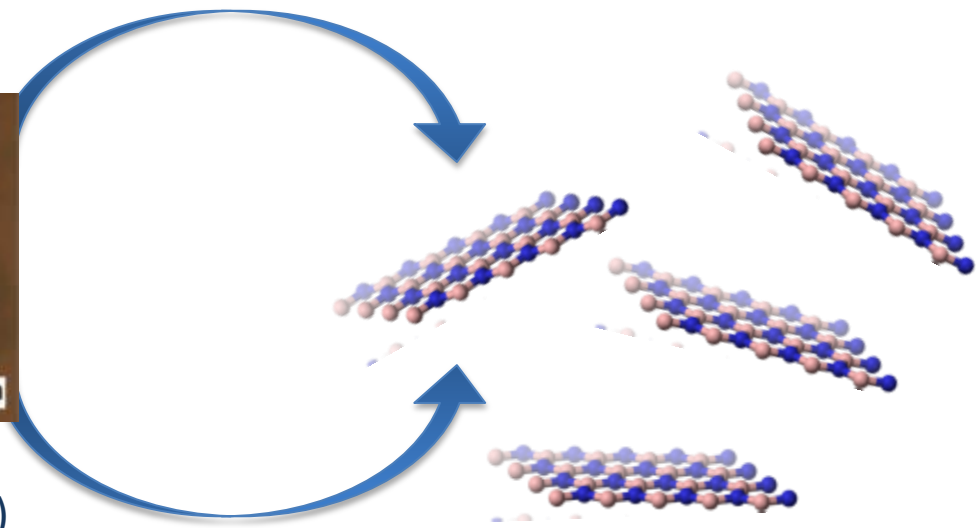
mechanical exfoliation  
using the tape method



well-crystallized  
h-BN bulk sample  
(pellet)



flake  
(single crystal)



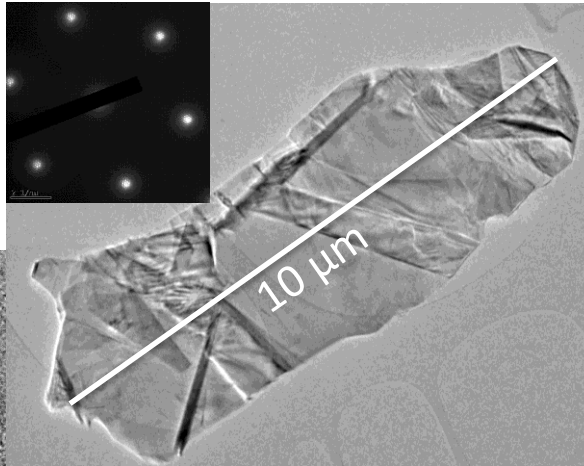
BNNSs

« chemical » exfoliation  
in a solvent under sonication

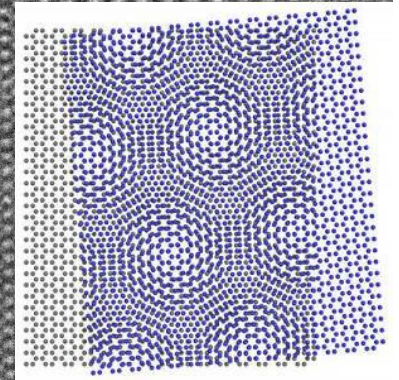
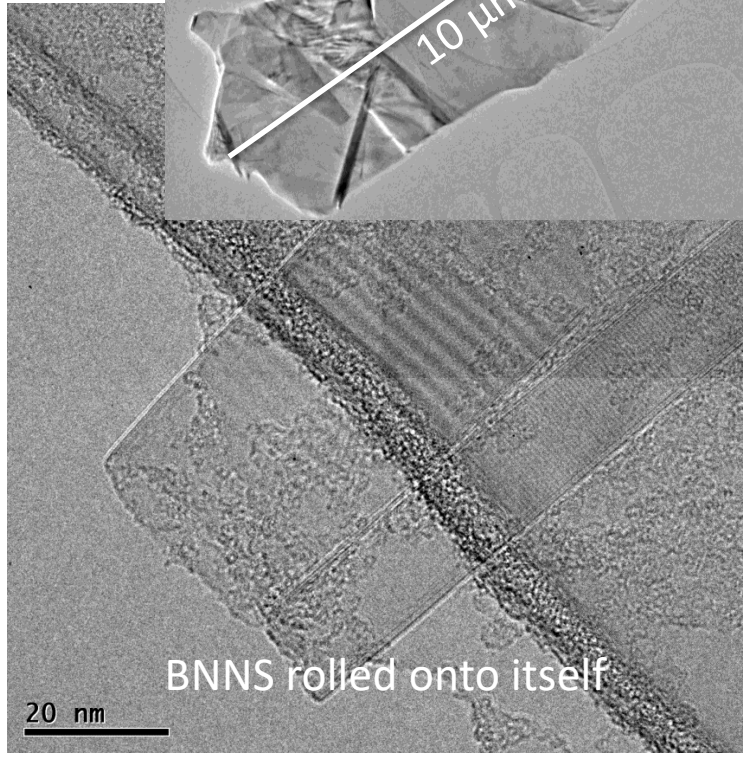
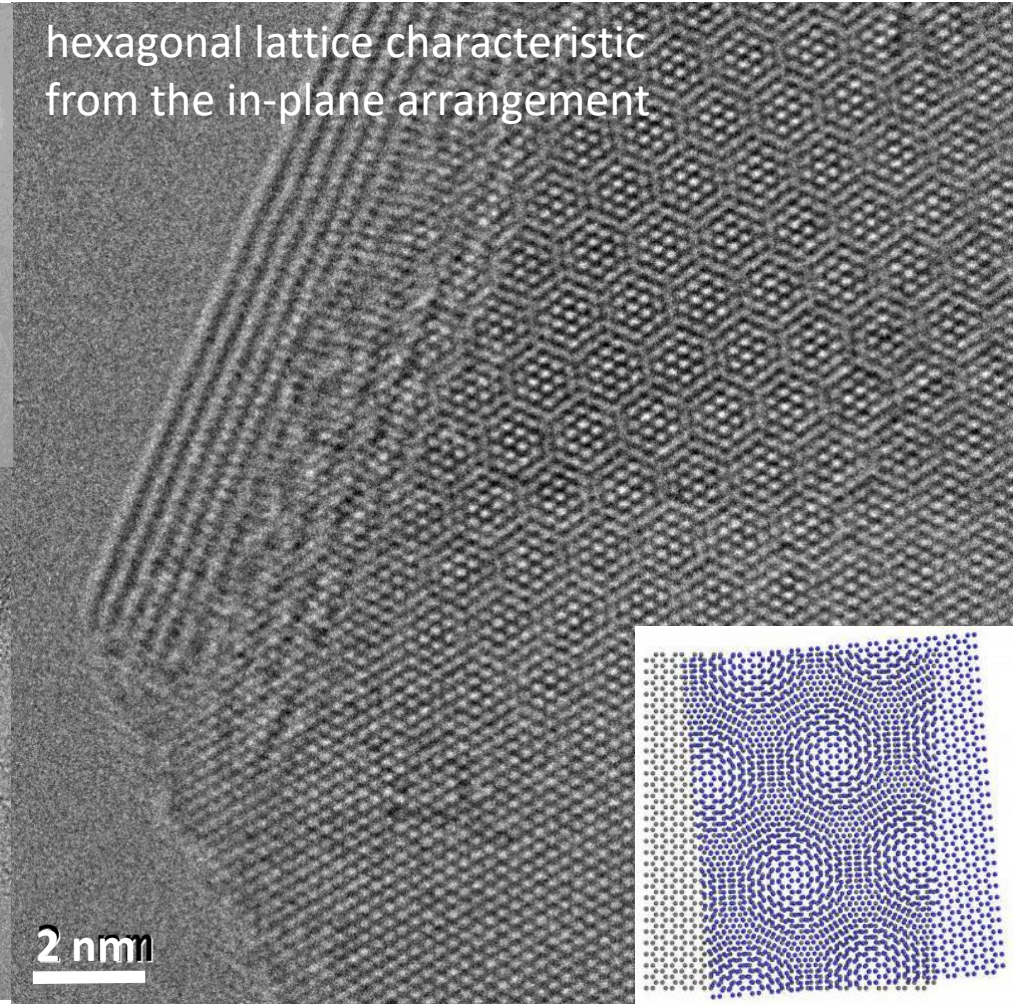


# Characterization of the BNNSs

TEM



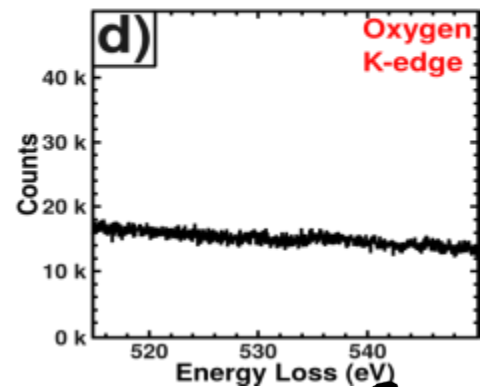
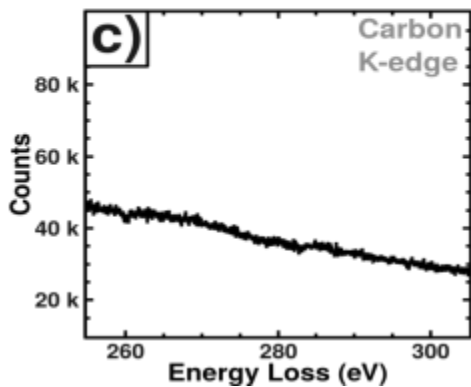
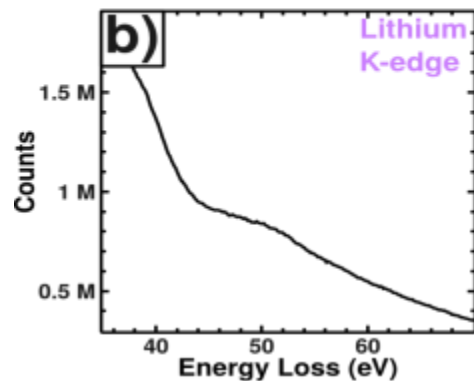
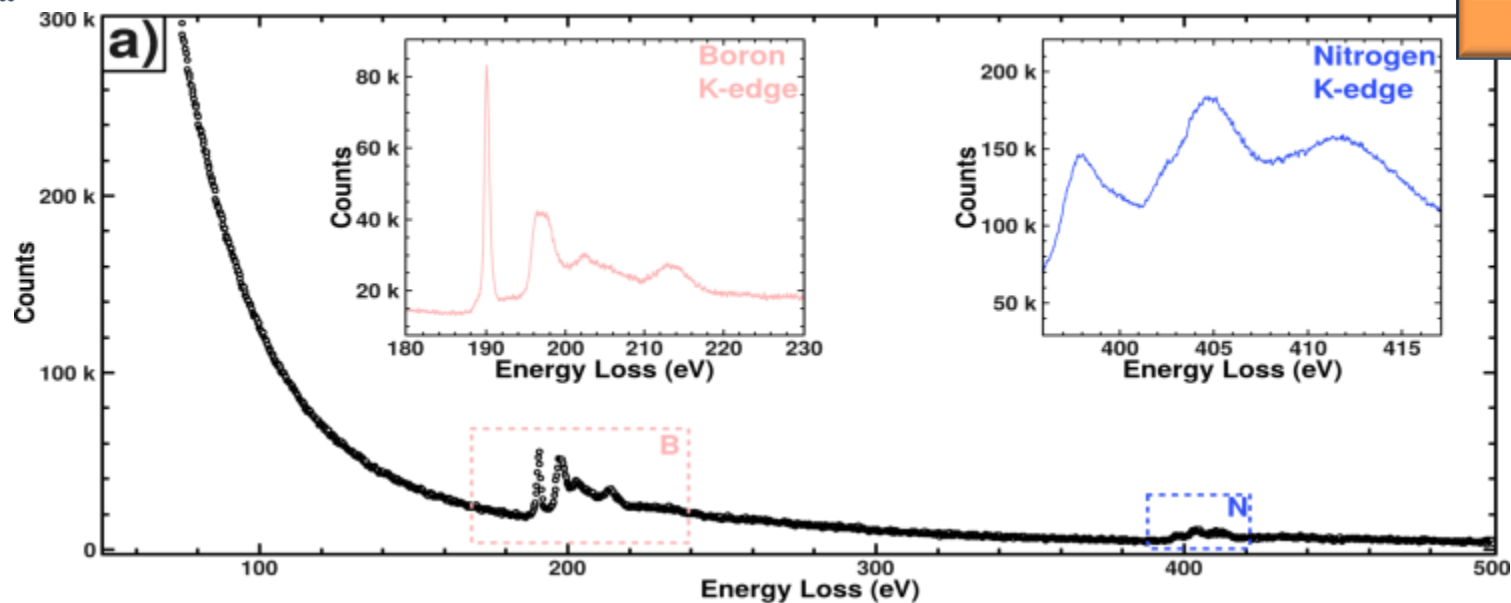
hexagonal lattice characteristic from the in-plane arrangement





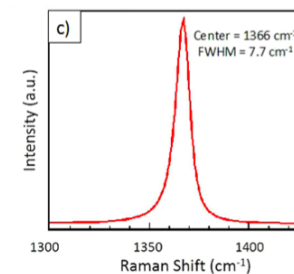
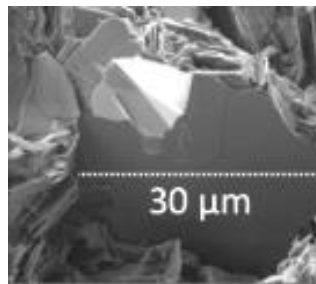
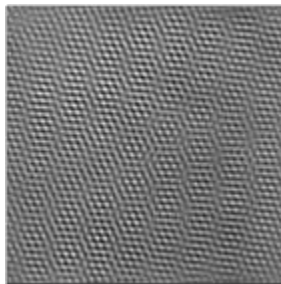
# Characterization of the BNNSs

EELS



## Conclusions & Outline

- BNNS as ideal candidate for graphene substrate / encapsulating layer / 2D layers staking
- Increase of the crystallinity by adding  $\text{Li}_3\text{N}$ 
  - Higher crystallinity at lower temperature
- Interest in combining the PDCs route & the SPS to get h-BN large single crystals
- After exfoliation, large (tens of  $\mu\text{m}$ ) and defect-free BNNSs obtained



*S. Yuan et al. Crystals, 6, 55 (2016)*





- Yangdi Li
- Bérangère Toury
- Sheng Yuan



- Philippe Steyer
- Vincent Garnier



**Thank you for your attention !**

- Annick Loiseau
- Alexandre Plaud
- Julien Barjon



**GRAPHENE  
FLAGSHIP**

